Probing magnetic symmetry in antiferromagnetic Fe₄Nb₂O₉ single crystals by linear magnetoelectric tensor

Jing Zhang,¹ Na Su,² Xinrun Mi,¹ Maocai Pi,¹ Haidong Zhou,³ Jinguang Cheng⁰,² and Yisheng Chai¹,*

¹Low Temperature Physics Laboratory, The Center of Quantum Materials and Devices, College of Physics, Chongqing University,

Chongqing 401331, China

²Beijing National Laboratory for Condensed Matter Physics and Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China ³Department of Physics and Astronomy, University of Tennessee, Knoxville, Tennessee 37996, USA

(Received 29 January 2021; accepted 29 March 2021; published 8 April 2021)

In the present study, we investigated magnetodielectric, magnetoelectric (ME), and angular-dependent polarization in single-crystal Fe₄Nb₂O₉. The magnetodielectric effects in ε_x (x//[100]), ε_y (y//[120]), and ε_z (z//[001]) are found to be significant only around $T_N \approx 95$ K when magnetic fields are applied along three orthogonal x-, y- (y//[120]), and z directions. The finite polarization P_x , P_y , and P_z of 70, 100, and 30 μ C/m², respectively, can be induced in the antiferromagnetic phase when a finite magnetic field up to 9 T is applied along the three orthogonal directions. The angular-dependent polarization measurements verify the dominating linear ME effects below T_N . From the above experimental results, a linear ME tensor a_{ij} with all nine nonzero components can be inferred, demonstrating a much lower magnetic point group of -1' for the canted antiferromagnetic configuration.

DOI: 10.1103/PhysRevB.103.L140401

I. INTRODUCTION

The direct linear magnetoelectric (ME) effect, where the electric polarization P is induced in proportion by an external magnetic field H [1,2], is expressed as $P_i = \alpha_{ij}H_j$ (a_{ij} is linear ME tensor where i and j run over all the Cartesian coordinates, x, y, z) [3,4]. It has attracted great interest due to the potential application for low power-consumption sensor devices and novel physical properties in solids [3,5-11]. This effect can exist in solids with long-range magnetic order, which breaks the space inversion and time-reversal symmetry simultaneously, like the time-honored Cr₂O₃ [12] with antiferromagnetic (AFM) order. Below $T_{\rm N} = 307$ K, Cr_2O_3 belongs to the magnetic point group -3'm'. From the Newmann principle, the nonzero ME tensor components are α_{xx} , α_{yy} , and α_{zz} . Only 58 out of 90 magnetic point groups allow the nonzero α_{ii} [13]. In this regard, symmetry analysis contributes to the searching for the linear ME effects in magnetic materials [1].

Recently, the investigation of $A_4Nb_2O_9$ (A = Mn, Co, Fe) has drawn strong attention as a potential ME material system, which belongs to the corundum-type structure firstly found by Bertaut *et al.* [14,15]. They crystallize in a trigonal crystal structure with the space group $P-3c_1$, a derivative of the $\alpha-Al_2O_3$ -type structure [14,16]. Two kinds of A atoms, A1 and A2, are located at 4d sites while Nb and two O atoms are located at 4c, 6g, and 12f Wyckoff positions, respectively, as shown in Fig. 1(a). The magnetic structures of both Mn and Co cases in powered samples were first determined to be a collinear AFM configuration along the c axis [14]. Their magnetic point group is -3'm' too, and has been confirmed in experiments [2,3,5,6].

2469-9950/2021/103(14)/L140401(5)

2'/m should be [18]

respectively.

However, $Fe_4Nb_2O_9$ shows an in-plane AFM configuration. The magnetic cation Fe^{2+} has a $3d^6$ configuration

with S = 2 and is found to be ordered at $T_N \approx 90$ K (94 K,

97 K) by magnetic susceptibility [17–19]. It experiences an-

other structure phase transition around 77 K (80 K, 70 K), as

manifested by an anomaly in dielectric constant and verified

by powder neutron-diffraction experiment [18]. Below 77 K,

its space group is lowered to C2/c (point group of 2/m), which

is not presented in the Co and Mn cases. In Ref. [18], the

magnetic structure just below T_N is described by collinearly

arranged moments in the *ab* plane parallel to either x//[100]

or y/[120] directions, which can be assigned in either 2/m'

or 2'/m magnetic point group, as shown in Figs. 1(b) and 1(c),

respectively. The matrix forms of the ME tensor for 2/m' and

 $\alpha_{ij} = \begin{bmatrix} \alpha_{xx} & 0 & \alpha_{xz} \\ 0 & \alpha_{yy} & 0 \\ \alpha_{zx} & 0 & \alpha_{zz} \end{bmatrix} \text{ and } \alpha_{ij} = \begin{bmatrix} 0 & \alpha_{xy} & 0 \\ \alpha_{yx} & 0 & \alpha_{yz} \\ 0 & \alpha_{zy} & 0 \end{bmatrix},$

Later on, a single-crystal neutron-diffraction technique

with refined magnetic structure below T_N revealed that the

adjacent Fe1-Fe2 spin pairs along the z direction canted

about 5.81 $^{\circ}$ in the plane, as shown in Figs. 1(d) and 1(e). The

(1)

^{*}yschai@cqu.edu.cn



FIG. 1. (a) Schematic crystal structure of $Fe_4Nb_2O_9$. (b)–(d) Top view of magnetic structures with spin along *y*, *x*, and canted directions, respectively. (e) Side view of spin configuration in (d) with space inversions centers (marked as X) in Fe1 layers.

magnetic structure and related point group, one has to measure all nine α_{ii} components of this compound.

In this paper, we investigate the dielectric, pyroelectric, and angle-dependent polarization under different external Hin two Fe₄Nb₂O₉ single crystals. We found that Fe₄Nb₂O₉ exhibits (i) anisotropic magnetodielectric effects, consistent with Ref. [19], (ii) nine nonzero components in linear ME tensor α_{ij} reflected by *H*-induced polarization below 95 K, and (iii) polarization with sinusoidal oscillation induced by rotating magnetic field below T_N for $\mu_0 H = 6$ T. All these results show that Fe₄Nb₂O₉ is a linear ME material with magnetic point group -1' by symmetry analysis.

II. METHODS

The single crystals of Fe₄Nb₂O₉ were grown from the traveling solvent floating-zone technique [19]. For electrical measurements, two single crystals were cut into a rectangular shape of $0.6 \times 1.4 \times 1.08 \text{ mm}^3$ and $0.9 \times 2.52 \times 0.42 \text{ mm}^3$, with the largest surface normal to the x/[100] and z/[001]directions, respectively. Gold electrodes were deposited onto these faces for electric measurements. Relative dielectric constant (ε) measurements were performed at 100 kHz using an Agilent E4980A Precision LCR meter. Before the pyroelectric current (I) measurements, the specimen went through the ME annealing procedure from 99 K. For $+H_p$ and $\pm E$ annealing condition, the sample was kept electrically poled with $+\mu_0 H_p = 0, 1, 2, 4, 9 \text{ T}, \pm E_1 = \pm 183 \text{ kV/m}$ (//x or //z) and $\pm E_2 = \pm 79.4 \text{ kV/m}$ (//y, due to much longer geometry along this direction) with external H down to 77 K. Then, $\pm E_1$ or $\pm E_2$ was turned off and the measurements were performed by sweeping T to cross the boundary between ferroelectric and paraelectric region, i.e., from 77 to 99 K on warming at a rate of 1 K/min using a Keithley 6517B electrometer. The temperature-dependent pyroelectric currents, I_x , I_y , and I_z along the x, y, and z directions, respectively, were measured and integrated as a function of time to determine the P under selected H values and orientations. The linear ME tensor components are estimated from average 9 T data in each pair of $+H_p$ and $\pm E$ configurations. The temperature and the magnetic field were controlled using a Dynacool system with 9 T superconducting magnet (Quantum Design). Angulardependent polarization measurements were performed on a rotator (MultiField Tech) in the Dynacool system.

III. RESULTS AND DISCUSSION

A. Dielectric constant measurements

In previous works, it was revealed that Fe₄Nb₂O₉ undergoes an antiferromagnetic and a structure phase transition at $T_{\rm N} = 90 \,\text{K} (94 \,\text{K}, 97 \,\text{K})$ and 77 K (80 K, 70 K), respectively [17–19]. In order to investigate the magnetoelectric properties in the AFM phase below T_N , we investigate the T-dependent ε_i under selected H_i (i, j = x, y, and z) in two Fe₄Nb₂O₉ single crystals, as shown in Fig. 2. For ε_x in warming, it quickly increases below 88 K and becomes flat until $T_N =$ 95 K where a small kink was observed for $\mu_0 H = 0$ T. This is consistent with the reported behavior in Ref. [19] with 20 kHz. When an external magnetic field is applied along the x direction, no change of ε_x can be observed except a small temperature region around T_N (92 to 96 K). Up to 9 T, the small kink around $T_{\rm N}$ becomes a clear bump, as indicated in the inset of Fig. 2(a). Our data have a better resolution than those in Ref. [19] in which a magnetodielectric effect with the same configuration of E//H//x cannot be observed clearly. When an external magnetic field up to 9 T is applied along the y direction, the change of ε_x is almost negligible in the entire measured T region. However, for H//z configuration, weaker bump features around $T_{\rm N}$ can be discerned as shown in the inset of Fig. 2(c). No strong magnetodielectric effects can be discerned in other temperature regions. For T-dependent ε_y with H//x, y, and z, they are almost identical with that of ε_x . It is due to the very weak in-plane crystal anisotropy in hexagonal lattice.

For *T*-dependent ε_z with $\mu_0 H = 0$ T, ε_z increases monotonically from 75 to 99 K while no anomaly can be found around T_N , consistent with the previous report [19]. When an external magnetic field along the *x* direction is applied up to 9 T, a clear peak in ε_z is induced around T_N , as shown in Fig. 2(g), while ε_z in the entire *T* region is slightly enhanced with a small offset. For H//y configuration, the magnetodielectric behavior is very similar to that of H//xconfiguration. Note that the $\mu_0 H = 0$ T data are slightly different among three *H* configurations due to the reloading of the sample at room temperature. For H//z configuration, the



FIG. 2. The temperature-dependent relative dielectric constant of Fe₄Nb₂O₉ ε_x for (a) H//x, (b) H//y, and (c) H//z, ε_y for (d) H//x, (e) H//y, and (f) H//z, and ε_z for (g) H//x, (h) H//y, and (i) H//z. All the data are measured in warming. Insets of (a)–(f) and (i) show the data around 94 K.

magnetodielectric effect is weakest with a small bump around $T_{\rm N}$ up to 9 T. We can conclude that the clear magnetodielectric effects of Fe₄Nb₂O₉ around $T_{\rm N}$ strongly indicate the existence of ME effects in all nine configurations below 95 K, which is inconsistent with our previous study [18]. Note that the small feature around 88 K may indicate a nonmagnetic phase transition which requires further investigation.

B. Magnetoelectric effects

To investigate the ME effects in the AFM phase just below $T_{\rm N}$ in this compound, we performed a comprehensive T-dependent polarization measurement along the P//x, y, and z axis under different H orientations, as shown in Fig. 3. From the above configurations, we expect to probe nine components in the linear ME tensor α_{ii} , where 2'/m and 2/m'magnetic point groups allow conjugation of nonzero components in Eq. (1). Without bias H, no ferroelectricity can be observed along crystal x, y and z directions. Under finite $\mu_0 H$ up to 9 T, P_x , P_y , and P_z can be gradually induced up to about 70, 100, and $30 \,\mu\text{C/m^2}$, respectively, without saturation. Moreover, the induced P values can be fully reversed by reversing the poling *E*, consistent with Ref. [19], which indicates that Fe₄Nb₂O₉ is a magnetoelectrics in the AFM phase. Surprisingly, even though the magnetodielectric effects are negligible for H//z configuration for ε_v and ε_z , the induced P_{v} and P_{z} values are comparable to or even larger than that in H//x and y directions for 9 T data, which requires further investigations. P_x , P_y , and P_z values at 90 K show quasilinear dependence with magnetic field along all three orthogonal directions, as shown in the insets of Fig. 3, implying dominating linear ME effects. To further verify this, we performed a reversal of polarization by reversing magnetic-field experiments below T_N , as shown Fig. 4(e). The sample is first ME poled by +E and $\mu_0 H = +9$ T down to 77 K. Then it will be warmed up to 99 K and 9 T with measuring the pyroelectric currents I_x and I_z . Or, the magnetic field will sweep to -9 T at 77 K, and the sample is warmed up to 99 K and -9 T with measuring the pyroelectric currents. As shown in Figs. 4(a) and 4(c), the signs of I_x and I_z are reversed by reversing the $\mu_0 H//x$ and $\mu_0 H//z$ from 9 to -9 T. Accordingly, the P_x and P_z can be fully reversed, as shown in Figs. 4(b) and 4(d), respectively. The polarization reversal by magnetic field again points to the dominating linear or oddorder ME effects in this system and consistent with the similar previous report with P_z and $\mu_0 H//x$ configuration [19]. We will show below that the linear ME effects are dominating in this system. From the 9 T and 90 K data in Fig. 3, the linear tensor components of $\alpha_{xx}, \alpha_{xy}, \alpha_{xz}, \alpha_{yx}, \alpha_{yy}, \alpha_{yz}, \alpha_{zx}, \alpha_{zy}, \alpha_{zz}$ are estimated to be 8.9, 13.0, 9.0, 13.4, 13.6, 15.1, 4.5, 4.5, 4.4 ps/m, respectively, in absolute values, pointing to a much lower magnetic point group at 90 K. We have to point out that the P and related ME tensor components are almost fully saturated under poling E fields applied.

C. Angular-dependent polarization

To unambiguously distinguish between the linear and higher odd-order ME effects, we perform an angular (θ) dependence of polarization P_x under rotating *H* experiments in the x-z plane as an example. Linear, third and higher-order terms will result in a function of $\sin \theta$, $\sin 3\theta$, and $\sin 5\theta$... terms in P_x , respectively. Particularly, for the linear ME tensor, $P_x = \alpha_{xx}H_x + \alpha_{xz}H_z$. Here, the magnetic field



FIG. 3. The temperature-dependent P_x , P_y , and P_z under orthogonal *H* along the (a), (d), (g) *x* direction; (b), (e), (h) *y* direction; and (c), (f), (i) *z* direction, respectively. Three insets plot the *H*-dependent polarization values at 90 K calculated from (a)–(i).

 $\mu_0 H = 6 \text{ T}$ rotates from x to z directions, as shown in Figs. 4(f) and 4(g). We can clearly see that the angular-dependent P_x shows dominating sinusoidal behaviors with small hysteresis between θ increase and decrease runs. This is a direct proof of dominating linear ME effect in this system. In this configuration, $H_x = H \cos \theta$ and $H_z = H \sin \theta$, leading to

$$P_{x} = H(\alpha_{xx}\cos\theta + \alpha_{xz}\sin\theta) = H\sqrt{\left(\alpha_{xx}^{2} + \alpha_{xz}^{2}\right)}\cos(\theta - \theta_{0}),$$
(2)

where $\theta_0 = \arctan(\alpha_{xz}/\alpha_{xx})$. From Fig. 4(g), the small average offset value of $\theta_0 \approx -32^\circ$ can be deduced from the angle increase and decrease runs, indicating an $\frac{\alpha_{xx}}{\alpha_{xz}} \approx -1.6$ ratio value and being consistent with the calculated value of $|\frac{\alpha_{xx}}{\alpha_{xz}}| = 8.9/9.0 \approx 1.0$ from the data in Fig. 3. The negative

sign of θ_0 reveals that a_{xx} , and a_{xz} have the opposite sign in the same AFM domain. The hysteresis behaviors may be related to the magnetic anisotropy between in plane and out of plane.

D. Symmetry analysis

From the above results, $Fe_4Nb_2O_9$ is an intriguing linear ME material with all nine nonzero terms, which is in direct contrast to the reported magnetic point group of 2'/m or 2/m' which has at most four and five nonzero components, respectively. Recently, a noncollinear magnetic configuration with a 5.81° canting between the adjacent Fe1-Fe2 spin pairs along the *z* direction was proposed through single-crystal neutron diffraction, as shown in Fig. 1(e). Its magnetic point group was described as 2/m', while a more careful reexamination in



FIG. 4. The temperature-dependent pyroelectric currents I_x (a) and I_z (c) in H//x and H//z, respectively. The temperature-dependent polarizations P_x (b) and P_z (d) in H//x and H//z, respectively. (e) Schematic poling process of H reversing from 9 to -9 T. (f) Angular dependence of polarization under the rotating $\mu_0 H = 6$ T at T = 90 K. Schematic sample configuration is shown in the upper panel.

this paper indicates a -1' magnetic point group. In this configuration, space inversion center+time reversal in Fe1 layers is retained regardless of the spins tilting in this layer. However, the 2y operation in the middle of Fe2 layers is broken due to canting between Fe2 spins. The overall magnetic point group of Fe₄Nb₂O₉ is -1' with a linear ME matrix form:

$$\begin{bmatrix} \alpha_{xx} & \alpha_{xy} & \alpha_{xz} \\ \alpha_{yx} & \alpha_{yy} & \alpha_{yz} \\ \alpha_{zx} & \alpha_{zy} & \alpha_{zz} \end{bmatrix},$$
(3)

where all nine coefficients can be nonzero. This matrix form is consistent with our experimental results of nine nonzero components. There are only two magnetic point groups 1, and -1' can allow all nine ME components while the magnetic order at least preserves -1' symmetry. In this sense, we are confident to conclude a -1' magnetic point group at 90 K.

IV. CONCLUSION

In summary, we studied the magnetodielectric and magnetoelectric properties of Fe₄Nb₂O₉. The relative dielectric constants along x//[100], y//[120], and z//[001] show strong field-dependent behaviors around T_N . Accordingly, all nine linear ME tensor components are estimated to have finite values. From symmetry analysis, the magnetic point group of the antiferromagnetic configuration below T_N is determined to be -1' instead of 2/m'. Our work suggests that measurement of ME tensors is a practical approach to assist the determination of long-range magnetic order in the insulating magnetic materials.

ACKNOWLEDGMENTS

This work is supported by the Natural Science Foundation of China Grants No. 11674384, No. 11974065, No. 11904040, No. 12004056, No. 12025408, No. 11921004, and No. 11874400. This work has been supported by Chongqing Research Program of Basic Research and Frontier Technology, China (Grant No. cstc2020jcyj-msxmX0263), Fundamental Research Funds for the Central Universities, China (Grants No. 2020CDJQY-A056, No. 2020CDJ-LHZZ-010, and No. 2020CDJQY-Z006), and Projects of President Foundation of Chongqing University, China (Grant No. 2019CDXZWL002). We would like to thank G. W. Wang at Analytical and Testing Center of Chongqing University for her assistance. H.D.Z. acknowledges support from Grant No. NSF-DMR-2003117.

- [1] H. Schmid, Ferroelectrics 162, 317 (1994).
- [2] J.-P. Rivera, Eur. Phys. J. B 71, 299 (2009).
- [3] Y. Tokura, S. Seki, and N. Nagaosa, Rep. Prog. Phys. 77, 076501 (2014).
- [4] T. H. O'Dell, *The Electrodynamics of Magneto-Electric Media* (North-Holland, Amsterdam, 1970).
- [5] M. Fiebig, J. Phys. D 38, R123 (2005).
- [6] N. A. Spaldin, M. Fiebig, and M. Mostovoy, J. Phys.: Condens. Matter 20, 434203 (2008).
- [7] H. Schmid, J. Phys.: Condens. Matter 20, 434201 (2008).
- [8] J. Wang, J. B. Neaton, H. Zheng, V. Nagarajan, S. B. Ogale, B. Liu, D. Viehland, V. Vaithyanathan, D. G. Schlom, U. V. Waghmare, N. A. Spaldin, K. M. Rabe, M. Wuttig, and R. Ramesh, Science 299, 1719 (2003).
- [9] N. A. Spaldin and M. Fiebig, Science 309, 391 (2005).
- [10] W. Eerenstein, N. D. Mathur, and J. F. Scott, Nature (London) 442, 759 (2006).
- [11] S.-W. Cheong and M. Mostovoy, Nat. Mater. 6, 13 (2007).

- [12] T. R. McGuire, E. J. Scott, and F. H. Grannis, Phys. Rev. 102, 1000 (1956).
- [13] R. E. Newnham, Properties of Materials: Anisotropy, Symmetry, Structure (Oxford University Press, Oxford, 2005).
- [14] E. F. Bertaut, L. Corliss, F. Forrat, R. Aleonard, and R. Pauthenet, J. Phys. Chem. Solids 21, 234 (1961).
- [15] E. Fischer, G. Gorodetsky, and R. M. Hornreich, Solid State Commun. 10, 1127 (1972).
- [16] N. D. Khanh, N. Abe, H. Sagayama, A. Nakao, T. Hanashima, R. Kiyanagi, Y. Tokunaga, and T. Arima, Phys. Rev. B 93, 075117 (2016).
- [17] A. Maignan and C. Martin, Phys. Rev. B 97, 161106(R) (2018).
- [18] R. Jana, D. Sheptyakov, X. Ma, J. A. Alonso, M. Pi, A. Muñoz, Z. Liu, L. Zhao, N. Su, S. Jin, X. Ma, K. Sun, D. Chen, S. Dong, Y. Chai, S. Li, and J. Cheng, Phys. Rev. B **100**, 094109 (2019).
- [19] L. Ding, M. Lee, E. S. Choi, J. Zhang, Y. Wu, R. Sinclair, B. C. Chakoumakos, Y. S. Chai, H. D. Zhou, and H. B. Cao, Phys. Rev. Materials 4, 084403 (2020).